# **Engineering Neutron and X-Ray Diffraction**

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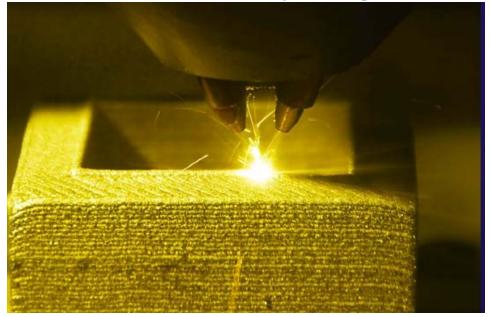
#### Additive Manufacture is of Particular Interest to the Nuclear Weapons Complex

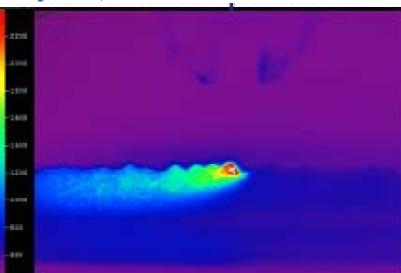
- AM is most efficient when applied to manufacture of small lots of complex geometries with minimal footprint and overhead investment...
  - ...once the process is qualified.
- AM can be agile and produce single parts for specific applications or experiments.
- Can we design specific functionality into components?
- But how do you qualify components made with a completely new technology?
- Our team is trying to play its roll in this task by using neutron and x-ray scattering techniques to monitor microstructure throughout lifetime of a component.
- So called Process/Structure/Properties/Performance relationship.



#### AM Inherently Spans Several Orders of Magnitude in Length and Time Scale.

- Quench rates following deposition can reach ~ $10^6 K/sec$ , pushing for phase quantification on µsec time scales.
- As material is subsequently added, temperature excursions can last as long as seconds.
- Post-build heat treatments last ~hour.
- Residual stresses build up over the duration of the build, hours.
- Their effect on performance can manifest in days-years.
- Dislocations, which control the material strength, exist on the atomic scale (Å).
- Phase topographies and grains are on ~μm scale.
- Powders are typically 10's of μm.
- Residual stresses are set by the length scale of the component, i.e. cm's.



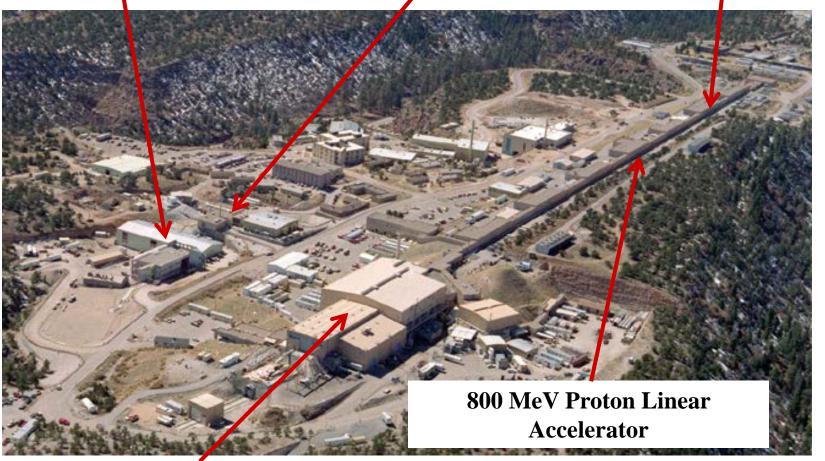




# Neutron Sources Do Not Sit on Desktops!

#### **Los Alamos Neutron Science Center: LANSCE**

Lujan Neutron Scattering Weapons Neutron Research Center Facility Isotope Production Facility

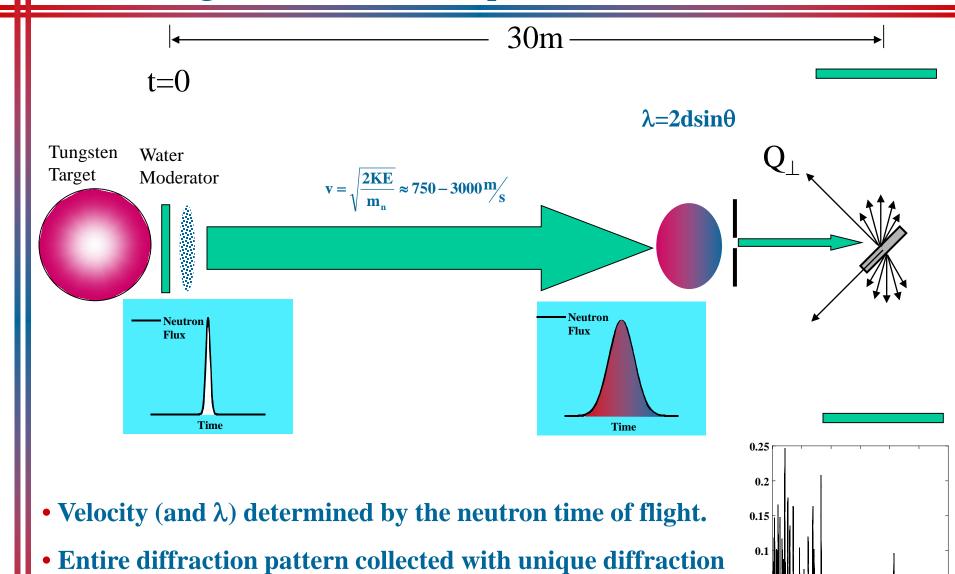


**Proton Radiography** 



# Time of Flight Diffraction at a Spallation Source

vector.

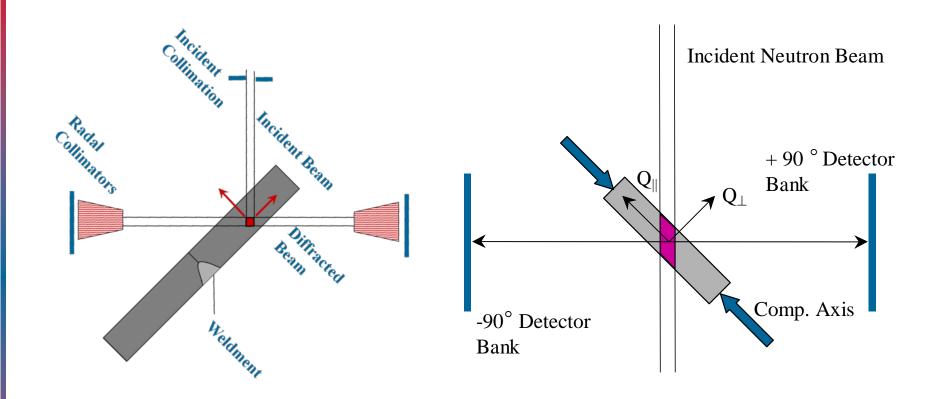


25

Time of Flight (msec)

30 35

# **Two Types of Measurements**



- Residual stress measurement:
  - You know the stiffness tensor and want to determine unknown stress
- Lattice response to external stimulus:
  - You control the stresses and you want to learn about the springs (bonds)

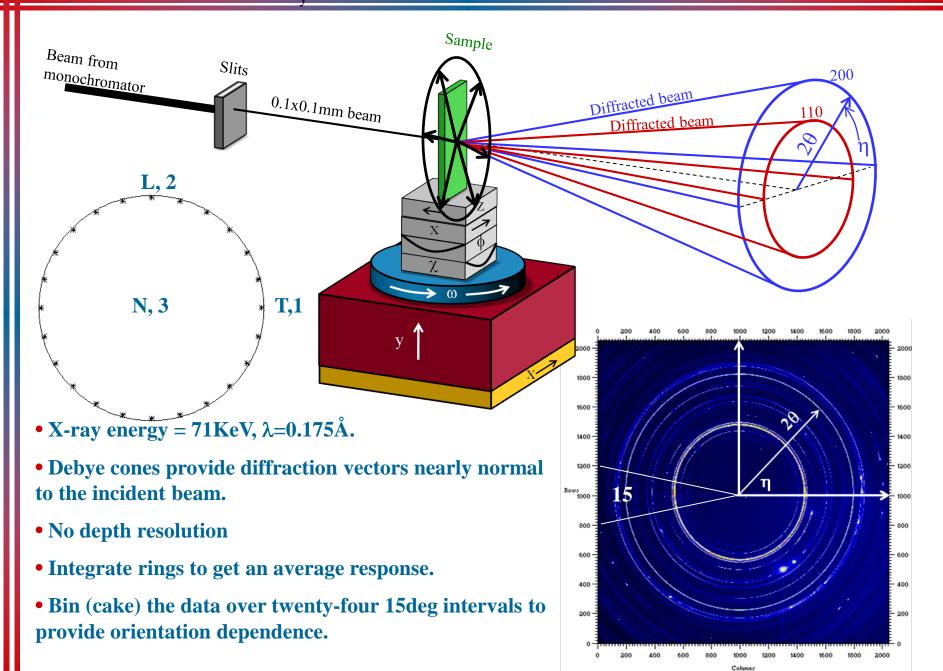


# **Lujan Scattering Team Utilizes Other Tools As Appropriate**

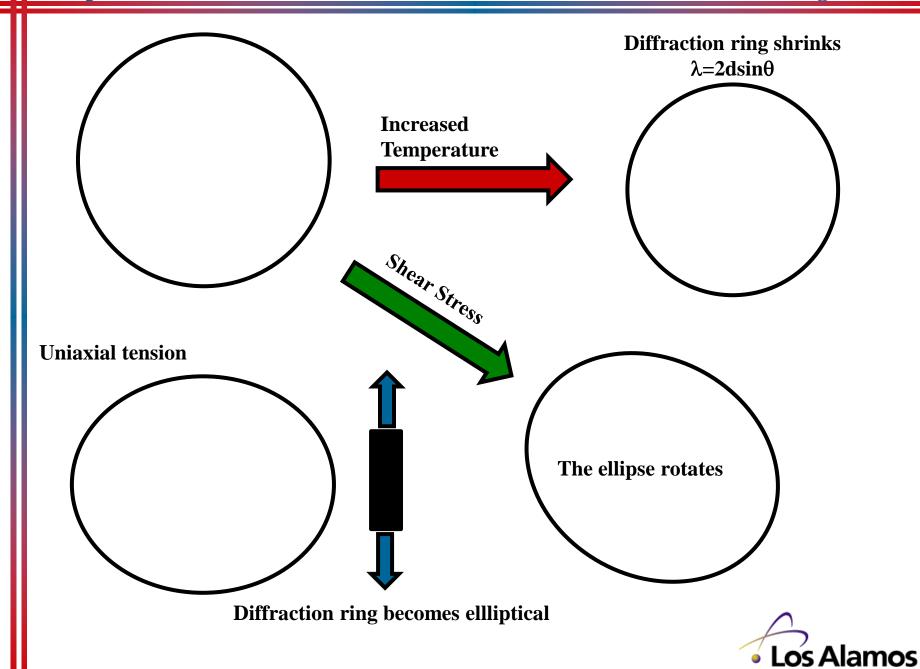




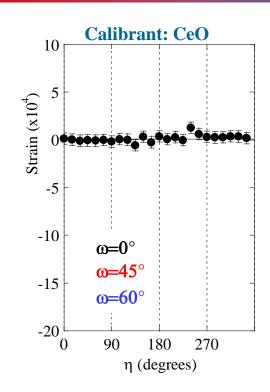
# High Energy X-Ray Diffraction Probes the Bulk of the Material

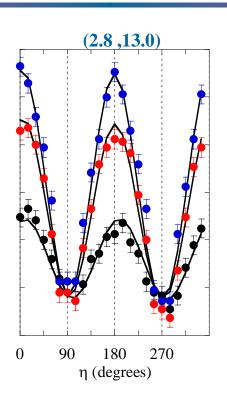


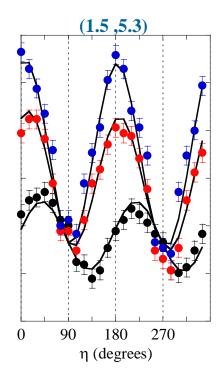
### **Temperature And Stress Have Distinct Effects on the Diffraction Rings**



### **Example of Residual Stress in U10Mo Thin Foil**



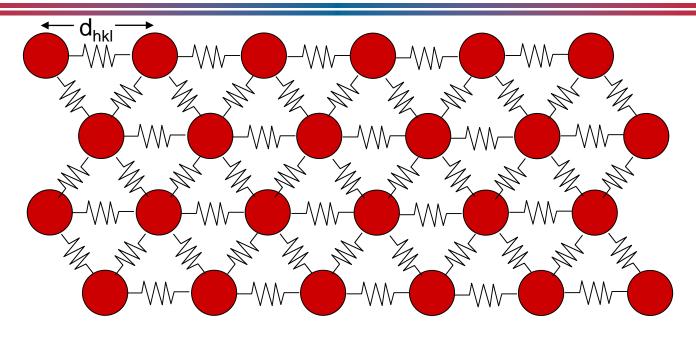




$$\varepsilon = \begin{cases} -8.1 & 1.4 & -0.5 \\ 1.4 & -16.0 & 1.0 \\ -0.5 & 1.0 & 12.8 \end{cases} \times 10^{-4} \pm \begin{cases} 0.3 & 0.3 & 0.3 \\ 0.3 & 0.2 & 0.3 \\ 0.3 & 0.3 & 0.4 \end{cases}$$



### We Use Diffraction Like A Bathroom Scale



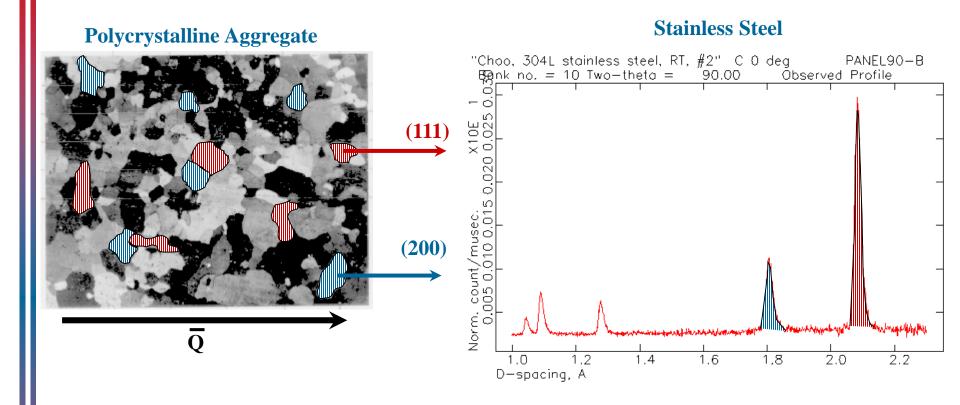
- We measure the spacing between atoms very precisely:  $\sim 1$  part in  $10^5$ .
- Calculate lattice strains (elastic) from change in atomic spacing.

$$\varepsilon = \frac{d_{hkl} - d_0}{d_0}$$

- ullet Generalized Hooke's Law  $: \sigma_{ij} = C_{ijkl} arepsilon_{kl}$
- It is important to note that the lattice strain is necessarily proportional to the stress on the grain, not the macroscopic stress.



# **Diffraction Separates Response of Grain Orientations**

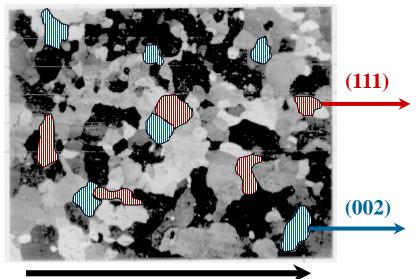


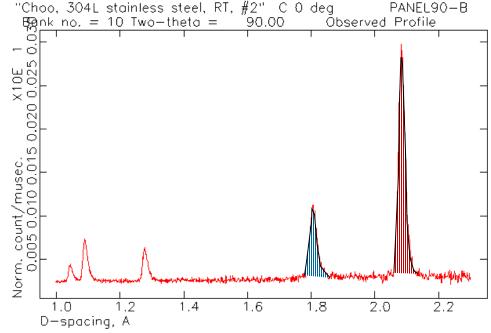
- Grains with plane normals parallel to the diffraction vector defined by the instrument geometry diffract into a detector.
- Each grain orientation (hkl), or phase, contributes to a distinct peak, given by the interplanar spacing.
- We explicitly make the assumption that a family of grains can be used to represent the macroscopic stress field.

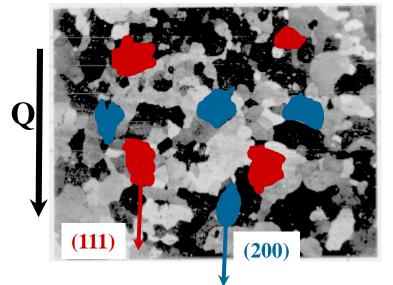
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# We Need to Step Back And Consider What Diffraction Tells Us

### **Polycrystalline Aggregate**



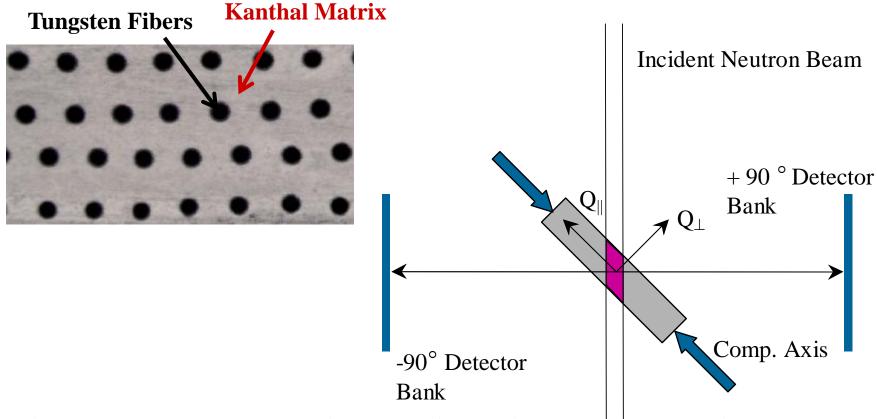




- Each diffraction peak samples grains with a common orientation.
- Many sample and crystal directions can be monitored.
- But cannot link peaks from different sample directions to unique grain sets.
- To calculate stress, I must assume that the differently oriented grains are under the same stress.

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#### **Lets Consider How a Composite Responds to Deformation**



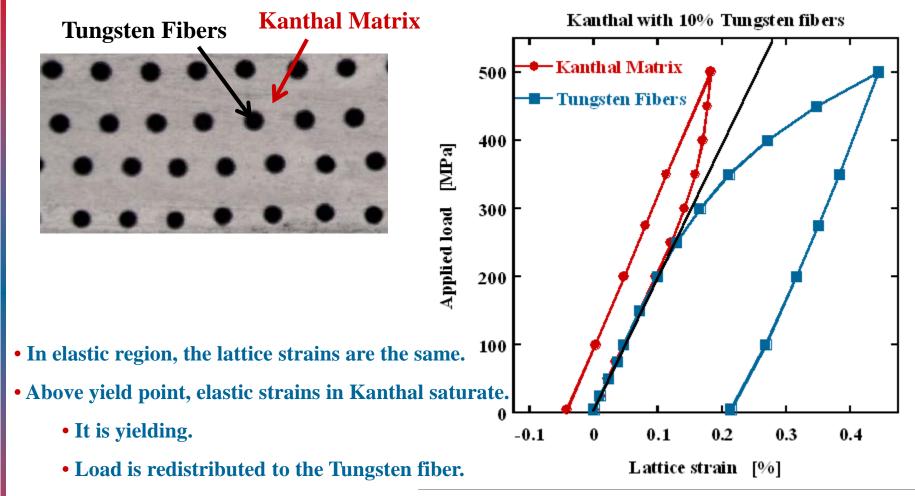
• Microstructure represents loading 2 constituents in parallel, total strains must be equal.

$$\varepsilon_T = \varepsilon_e + \varepsilon_p$$

- In elastic regime, lattice strains are equivalent.
- Once plasticity begins in one phase, the elastic lattice strains are no longer constrained to each other.

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# **Understand Anisotropy in Terms of a Composite**

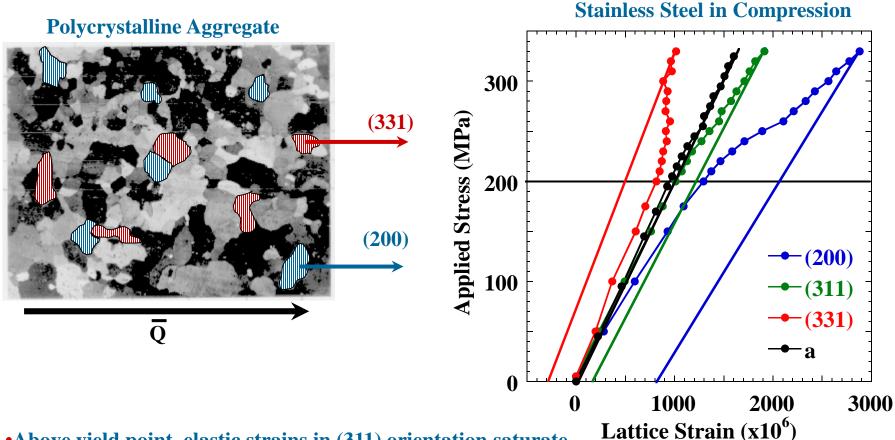


- This is how a composite is designed to work.
- With release of the macroscopic stress, there is a residual stress in each constituent.
  - The phases stresses are not representative of the macroscopic stress state.

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• However, a weighted average would be representative.

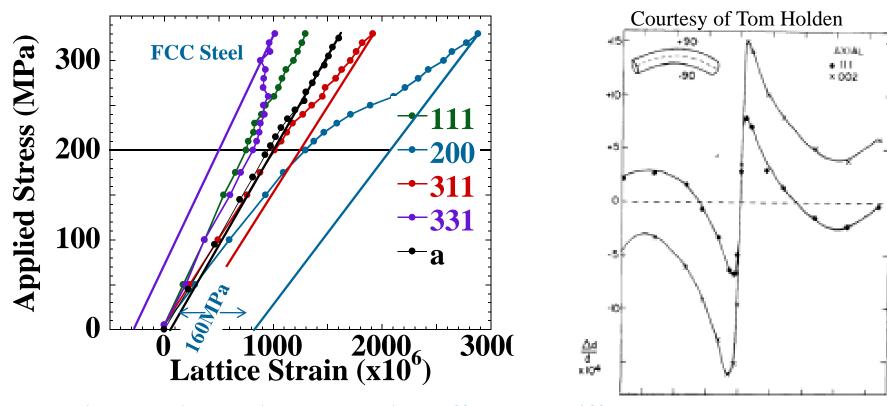
# Polycrystalline Samples: "The Mother of All Composites".



- •Above yield point, elastic strains in (311) orientation saturate.
  - Grains with (311) parallel to the load direction are yielding.
  - Load is redistributed to the (200) orientations.
- This is a general result seen in FCC metals.
  - Indicative of the specific deformation mechanism.



# There is a Long Time Strategy to Deal With Anisotropy



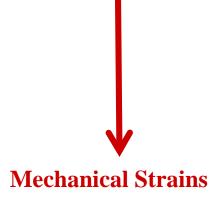
- Elastically anisotropic, but that just effects the stiffness you use.
- Plastic anisotropy will lead you to the wrong answer.
- Tom Holden has told us all not to use the (002) and (111) for FCC's.
- Solution: (311) reasonably represents the macroscopic stress state.
- The lattice parameter found from Rietveld refinement represents the bulk very well.

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# But What Can We Obtain From a Dynamic Measurement During Deposition?

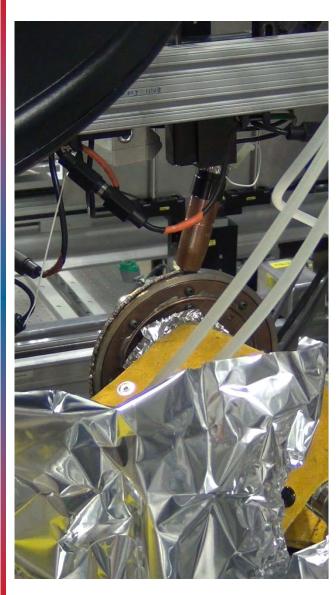
$$\begin{pmatrix} \varepsilon_{11} & \varepsilon_{12} & \varepsilon_{13} \\ \varepsilon_{21} & \varepsilon_{22} & \varepsilon_{23} \\ \varepsilon_{31} & \varepsilon_{32} & \varepsilon_{33} \end{pmatrix} = \begin{pmatrix} \varepsilon_h & 0 & 0 \\ 0 & \varepsilon_h & 0 \\ 0 & 0 & \varepsilon_h \end{pmatrix} + \begin{pmatrix} \varepsilon_{11} - \varepsilon_h & \varepsilon_{12} & \varepsilon_{13} \\ \varepsilon_{21} & \varepsilon_{22} - \varepsilon_h & \varepsilon_{23} \\ \varepsilon_{31} & \varepsilon_{32} & \varepsilon_{33} - \varepsilon_h \end{pmatrix}$$



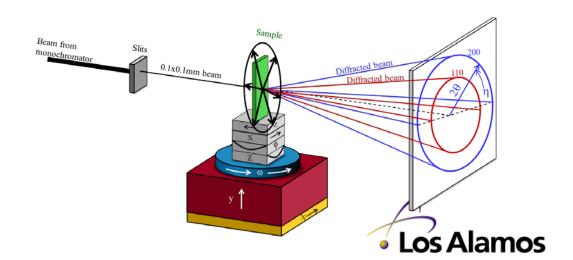




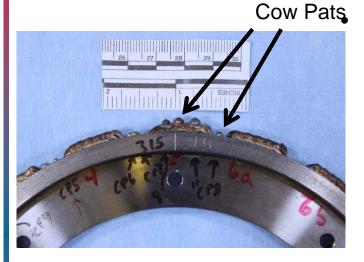
# Portable Additive Manufacturing Platform Developed for In-Situ Studies on X-ray and Proton Beamlines

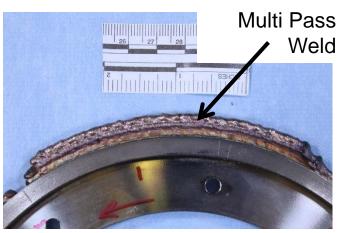


- Objective is to characterize the deposited material (and substrate) as the process is happening (in-situ).
- Cold Metal Transfer (CMT) weld process.
  - Like MIG but better controlled.
- 1ID @ the APS
  - E=71.67keV, λ=0.171Å
    - ~2 factors of e absorption.
  - Beam size 0.05-0.2mm for diffraction (could be smaller)
  - 1x2mm for imaging (largest FOV)
  - Rotating wheel speed ~1mm/sec



# 3 Types of In-Situ Measurements Were Completed on 1-ID





Ordered in increasing complexity and increasing information...

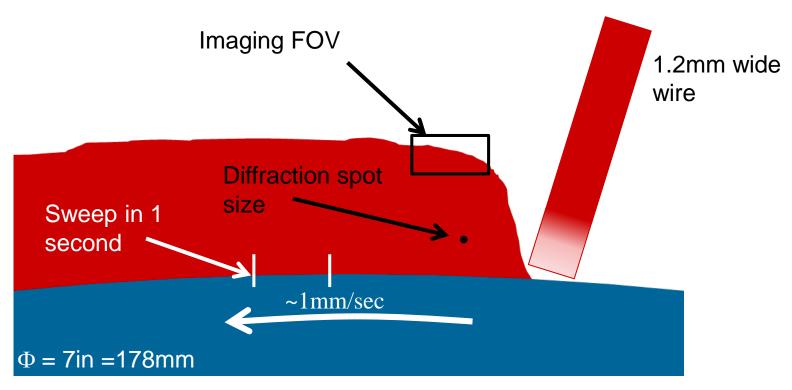
- Stationary substrate (wheel) with ~1 second deposition.
- Rotating wheel with incident x-ray spot fixed.
  - Multiple tests with different lateral beam positions.
- Rotating wheel while moving x-ray spot.
  - Multiple tests at different beam heights.

Others you will not see today...

- Post deposition residual stress measurements
- Measurements in wheel below deposition
- Laser re-melting of previously deposited material.



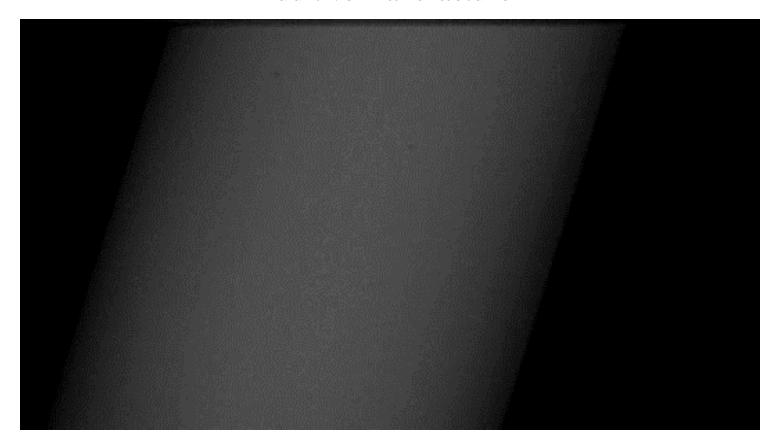
# Scale Relationships Between Sample and Probe Size



- Diffraction count time and spot size were dictated by statistical sampling of grains in integration volume.
  - Usually 1sec, sometimes 0.1sec (attenuators in beam).
- Imaging 0.1sec integration, limited by statistics.

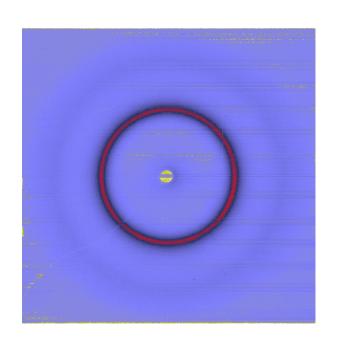


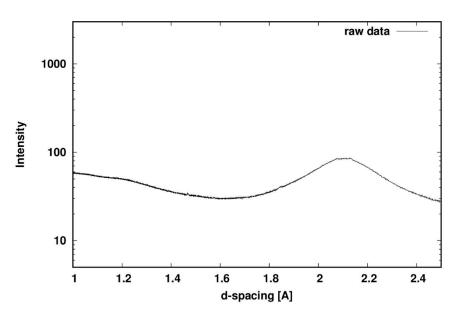
# **Additive Manufacture**

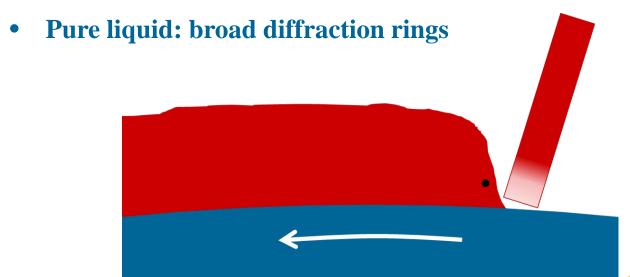






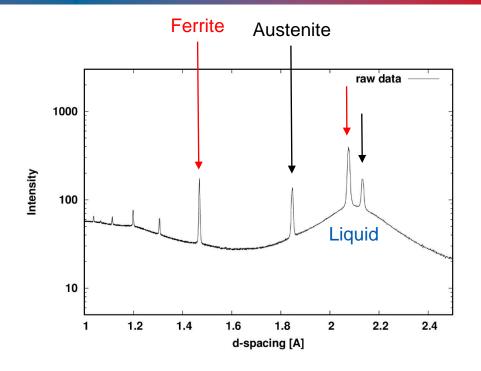








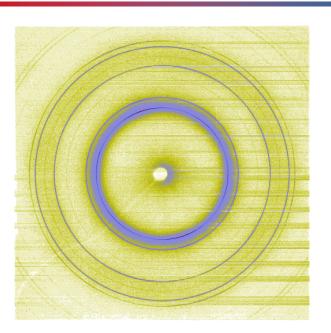


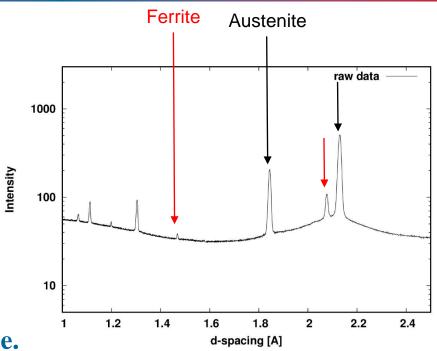


- Liquid with solid nuclei
- Integrated pattern shows sharp and broad peaks



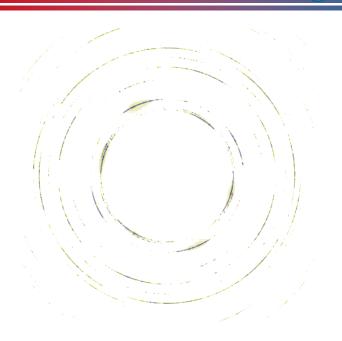
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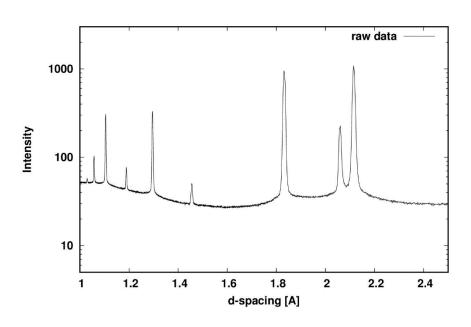




- Liquid and solid coexistence.
- Continuous, nearly uniform rings
  - Small grains, no texture.



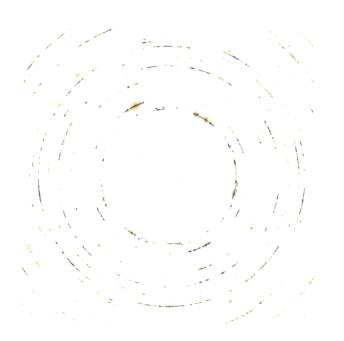


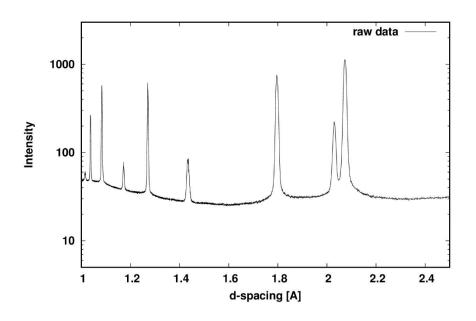


- Textured solid.
- Continuous but non-uniform rings
  - Small textured grains

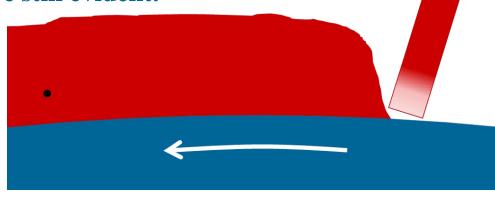






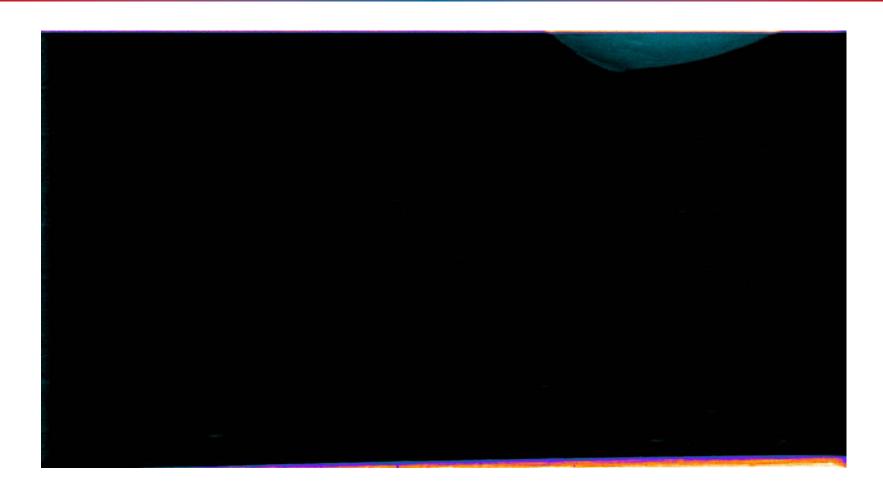


- "Spotty" rings.
- Solid with large, annealed grains.
- Texture still evident.





# The Dropping of a Cow Pattie

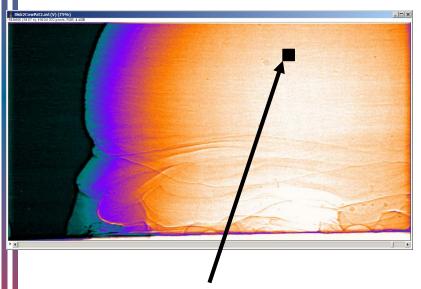




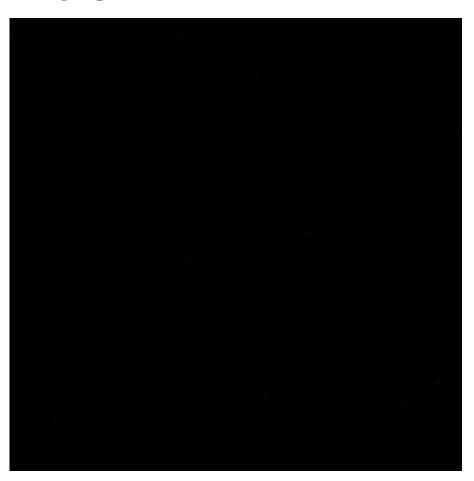
# Diffraction Used to Monitor the Microstructure of the Material During Deposition

#### In-situ diffraction probes phase evolution following deposition

- Liquid observed
- Pattern is spotty, grains are "large".



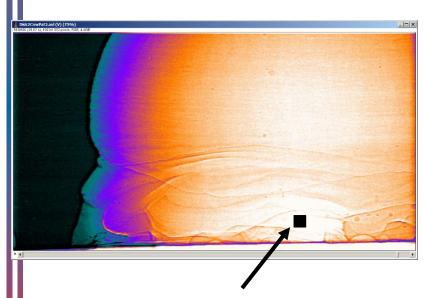
Diffraction spot 0.8mm from rim





#### Diffraction Used to Monitor the Microstructure of the Material During Deposition

- In-situ diffraction probes phase evolution following deposition
  - No liquid observed, rapid solidification.
  - Rings are more continuous, grains are smaller

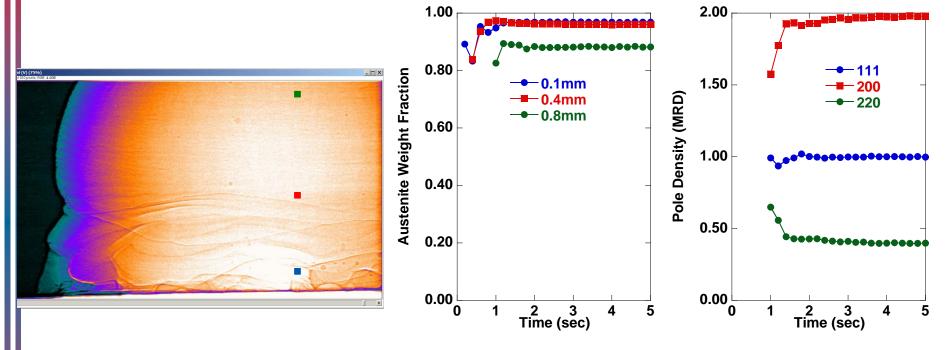


Diffraction spot 0.1mm from rim





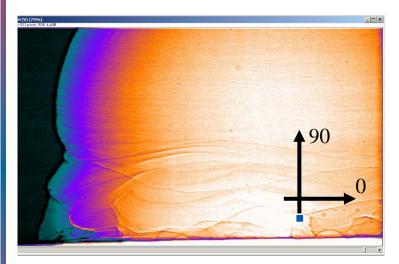
#### **Quantitative Microstructural Information is Extracted From the Diffraction Pattern**

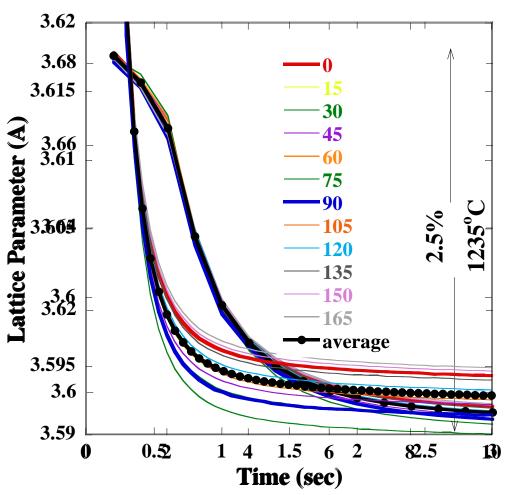


- Initially solidifies 80% Austenite (20% Ferrite)
  - Quickly stabilizes, but is a function of distance above wheel.
- Texture develops at highest measurement point, evolves in first ~0.5 second



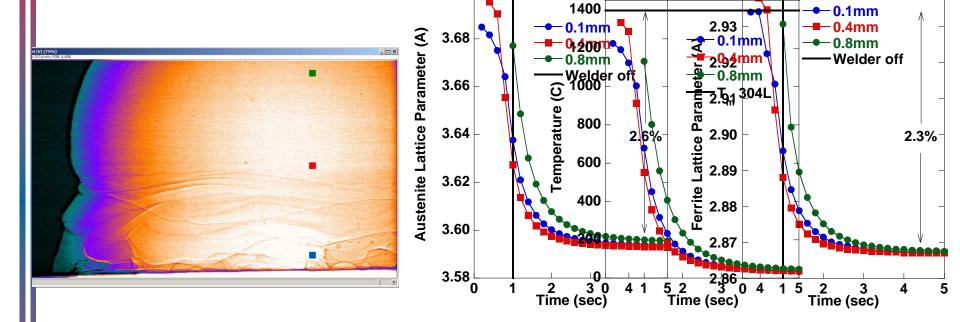
# Temperature and Stress Are Evident in Lattice Parameter Evolution







#### **Quantitative Microstructural Information is Extracted From the Diffraction Pattern**



3.70

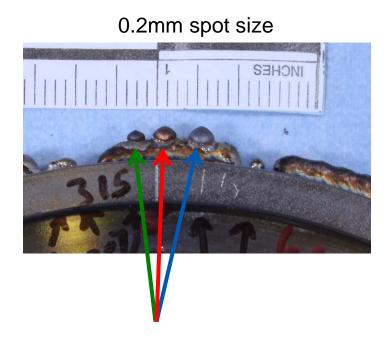
- Track the lattice parameter of both phases during and following the weld.
- **Lattice strain can be converted to temperature.** 
  - Are there other factors in the lattice parameter, e.g. chemistry or stress?

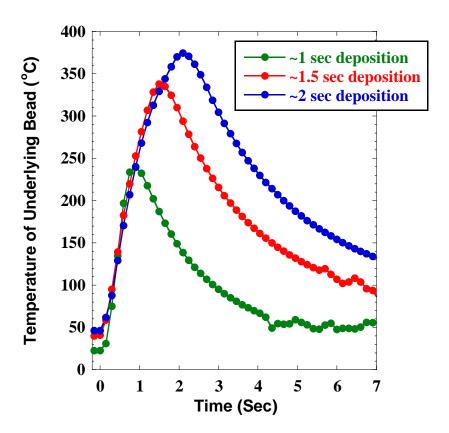


2.94

#### **Effect of Deposition on Previous Layer Observed**

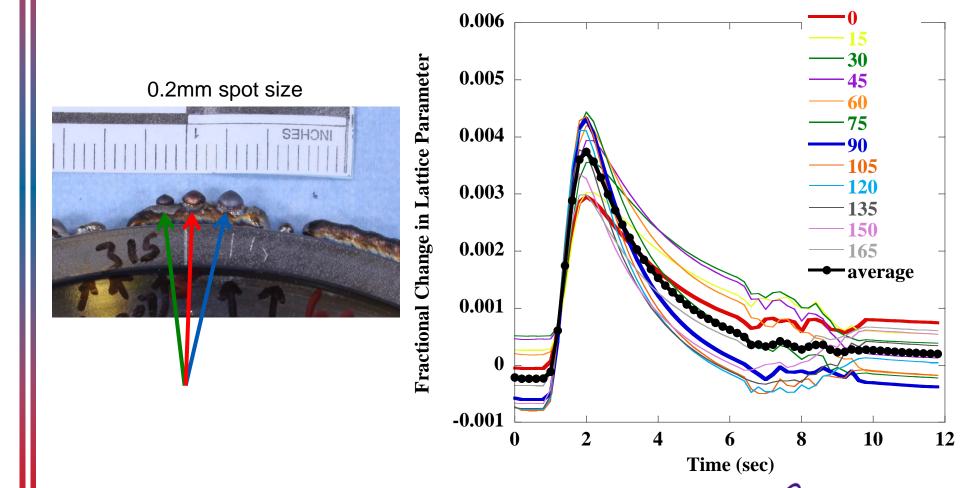
• Lattice parameter measured in-situ during welding used to determine temperature of material below deposition.







# **Effect of Deposition on Previous Layer Observed**

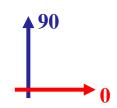


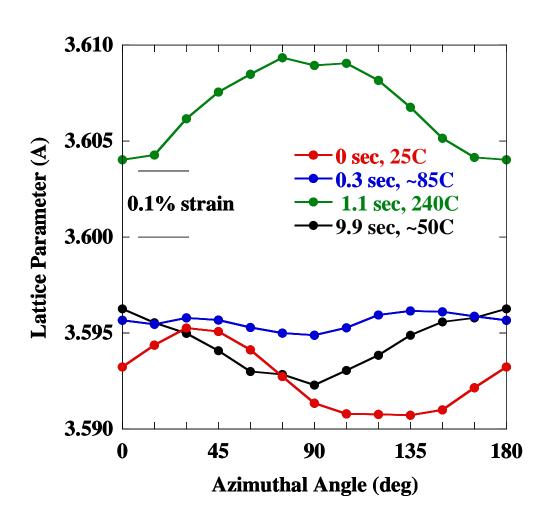
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#### **Evolution of Stress in Previous Layer is Evident**



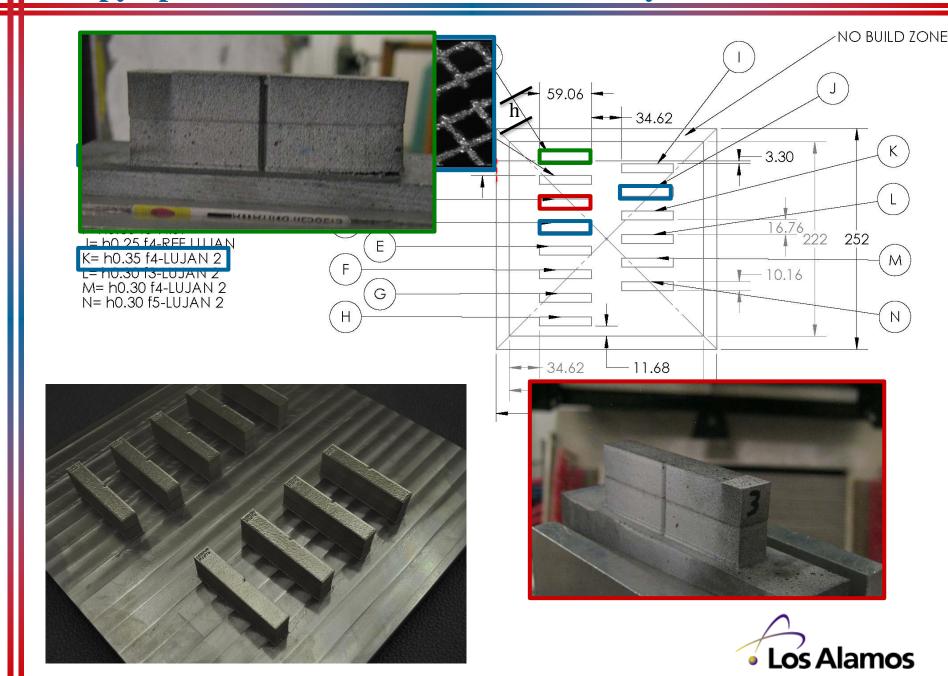




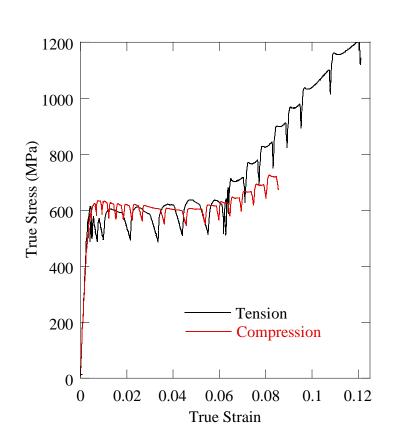




# Charpy Specimen Grown From GP-1 to Study PSPP



#### Measured Mechanical Response of the As-Built Material



- Composition (corresponds to 17-4)
  - Cr (15 17.5 wt-%)
  - Ni (3 5 wt-%)
  - Cu (3 5 wt-%)
  - Mn (max. 1 wt-%)
  - Si (max. 1 wt-%)
  - Mo (max. 0.5 wt-%)
  - Nb (0.15 0.45 wt-%)
  - C (max. 0.07 wt-%)
- Mechanical properties of parts
  - Ultimate tensile strength ~1000MPa
  - Yield strength ~550MPa
  - Young's modulus  $170 \pm 20$  GPa

Match EOS spec sheet (note non-typical flow curve).

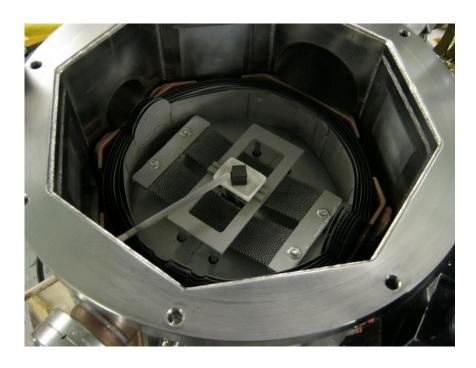
**Modulus of ~ 190GPa** 

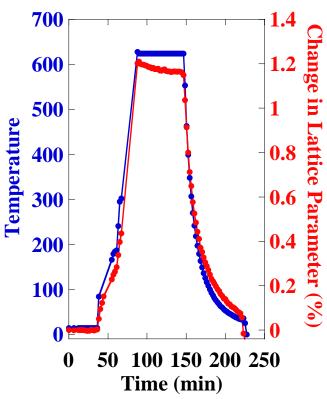
Yield strength ~ 600MPa

UTS greater than 1.2GPa, looks like ~1.5GPa



#### EOS Recommends a 1hr/650C "Stress Relief" Heat Treatment of GP-1 Builds

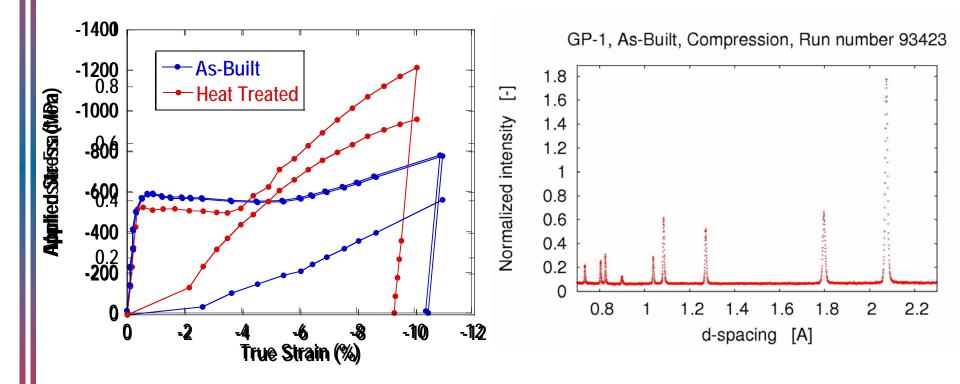




- Chemistry evolves during the heat treatment.
  - Analysis shows that interstitial nitrogen is lost during heat treatement.
- What effect does that process have on the structure/properties?



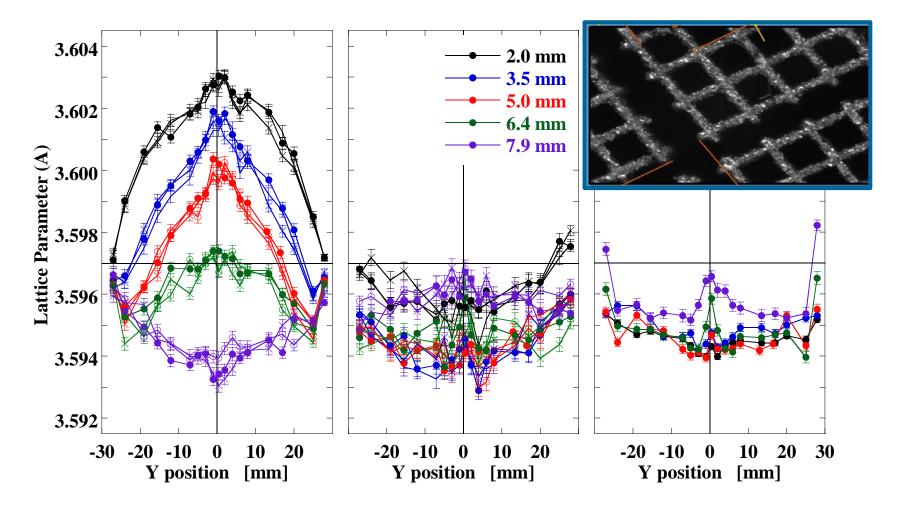
#### **Heat Treatment Results in Large Difference in Structure and Properties**



Loss of interstitial nitrogen increases propensity for martensitic transformation.

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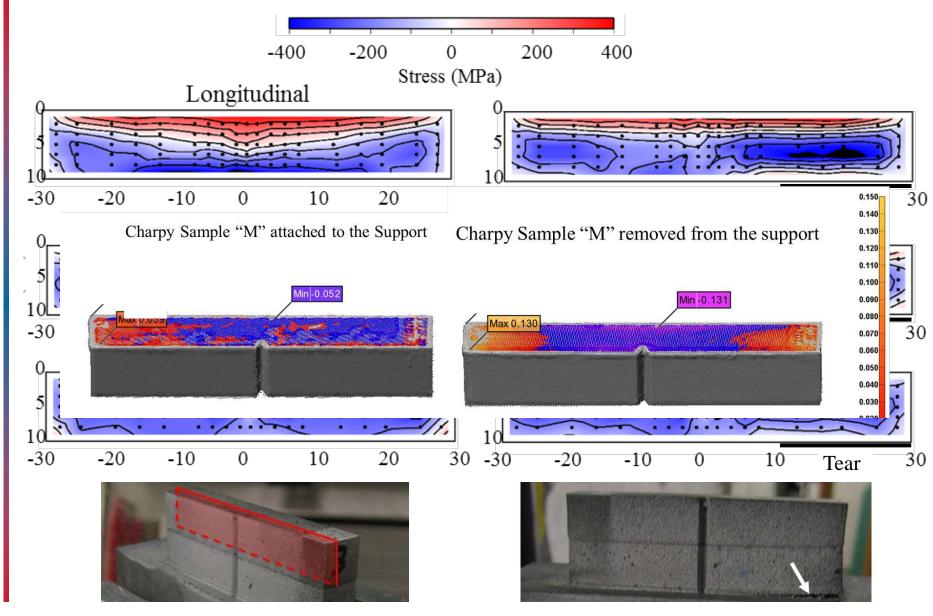
#### **Stress is Independent of Hatching and Fragmentation**



C: h = 0.25mm, f = 3mm D: h = 0.25mm, f = 5mm K: h = 0.35mm, f = 4mm

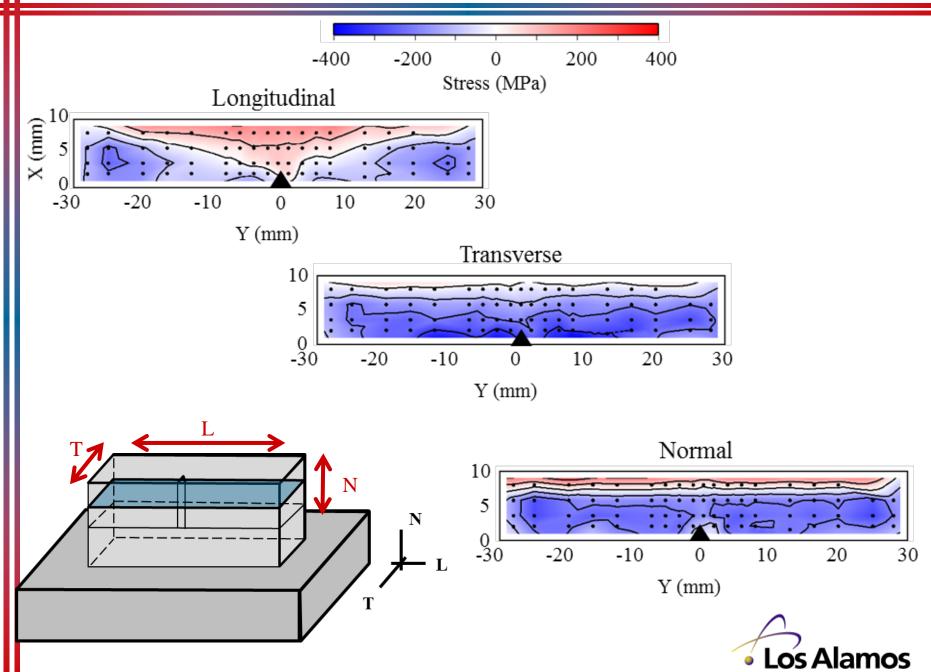


### **Stress is Strong and Predominantly Longitudinal**



Tear During Deposition Has a Strong Effect on Residual Stress

### Presence of Notch Results in an Asymmetry in the Stress Field

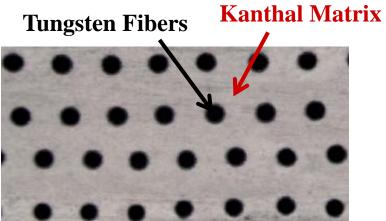


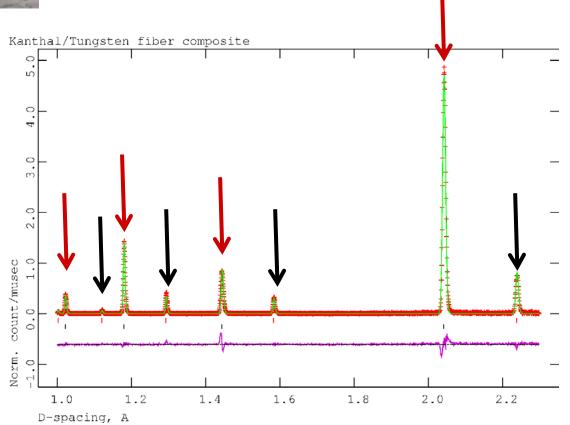
#### **Conclusions**

- Neutron and high energy synchrotron x-ray diffraction and imaging used to monitor microstructure and stress in engineering materials.
- Trying to understand Processing/Structure/Property/Performance relationship to that we can intelligently design and qualify novel materials processing.
  - Additive manufacture is a single example.
- X-ray diffraction has the promise for in-situ studies during deposition, with measurements in the millisecond time frame available today, microsecond or better in the future.
- Neutrons will be limited to post-processing and or operando or residual stress studies of material.
- Combined, they provide a window unto the entire PSPP relationship.



# **Diffraction Inherently Distinguishes Between Phases**





### **Diffracted Peak Intensity is Directly Related to Pole Density**

